# Computational modeling of SiC formation by C60 epitaxy on Si(1,1,1)-7x7 surface



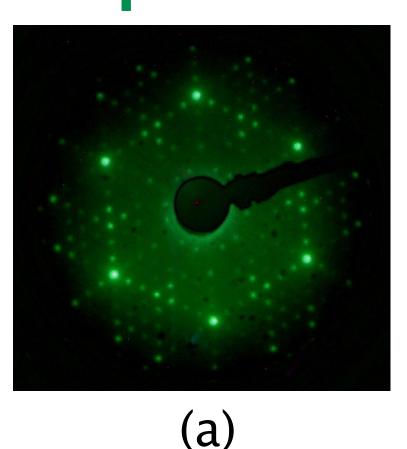
S. Taioli\*, G. Garberoglio\*, S. a Beccara\*, M. Dapor\*, S. Simonuccio, R. Verucchi‡, L. Aversa‡, M. Nardi‡, S. Iannotta‡, D. Alfè§

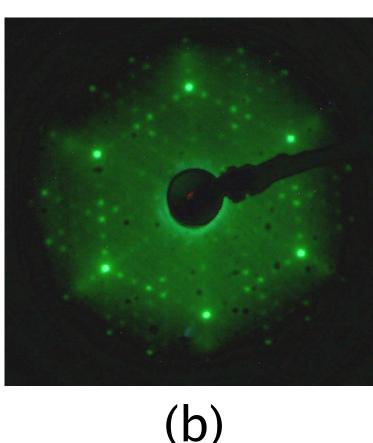
\*Laboratorio Interdisciplinare di Scienza Computazionale (LISC), Fondazione Bruno Kessler and University of Trento, Italy. <sup>o</sup>Dipartimento di Fisica, University of Camerino, Camerino, Italy ‡Institute of Materials for Electronics and Magnetism, IMEM-CNR, Trento, Italy. §Department of Earth Sciences, University College London, London, United Kingdom.

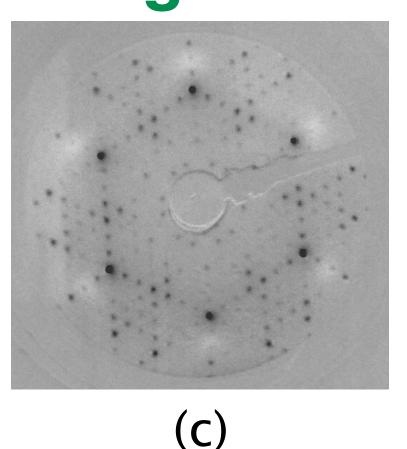
# Why SiC?

- SiC is used to replace Si in harsh working environments. More than 200 polytypes, wide band gap from 2.4 eV (3C) to 4.1 eV (2H).
- SiC is very rarely found in nature.
- The synthesis procedures are very expensive (high T, > 1300 K, lot of impurities).
- We seek for an inexpensive way of producing SiC at room temperature.

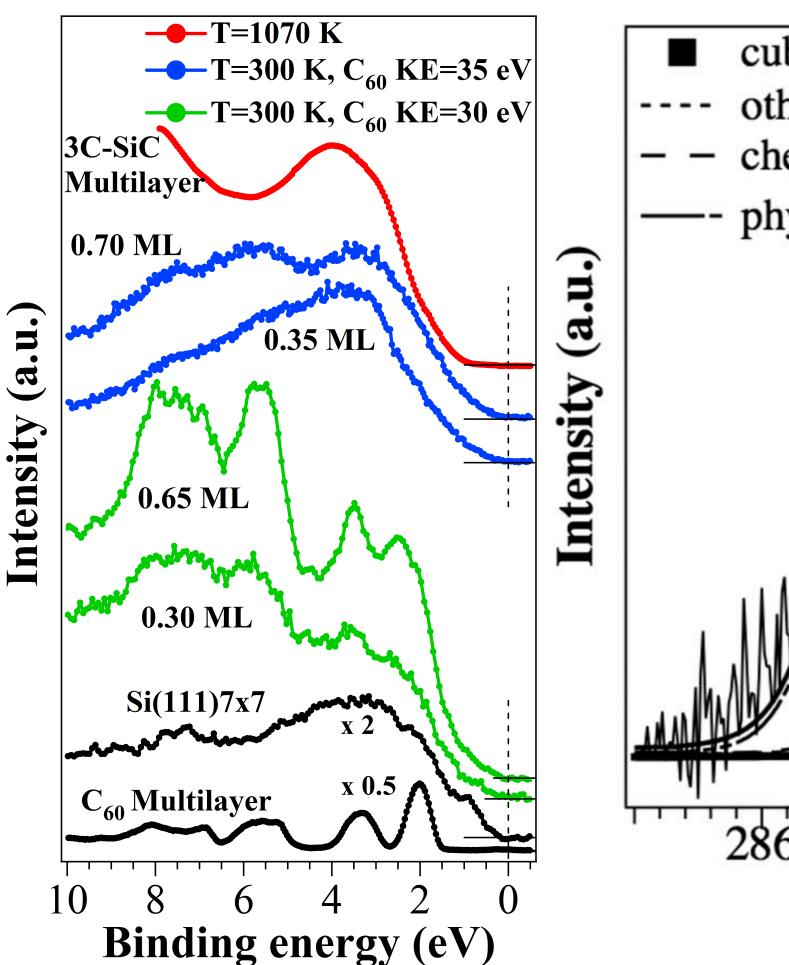
### **Experimental evidence of SiC growth**

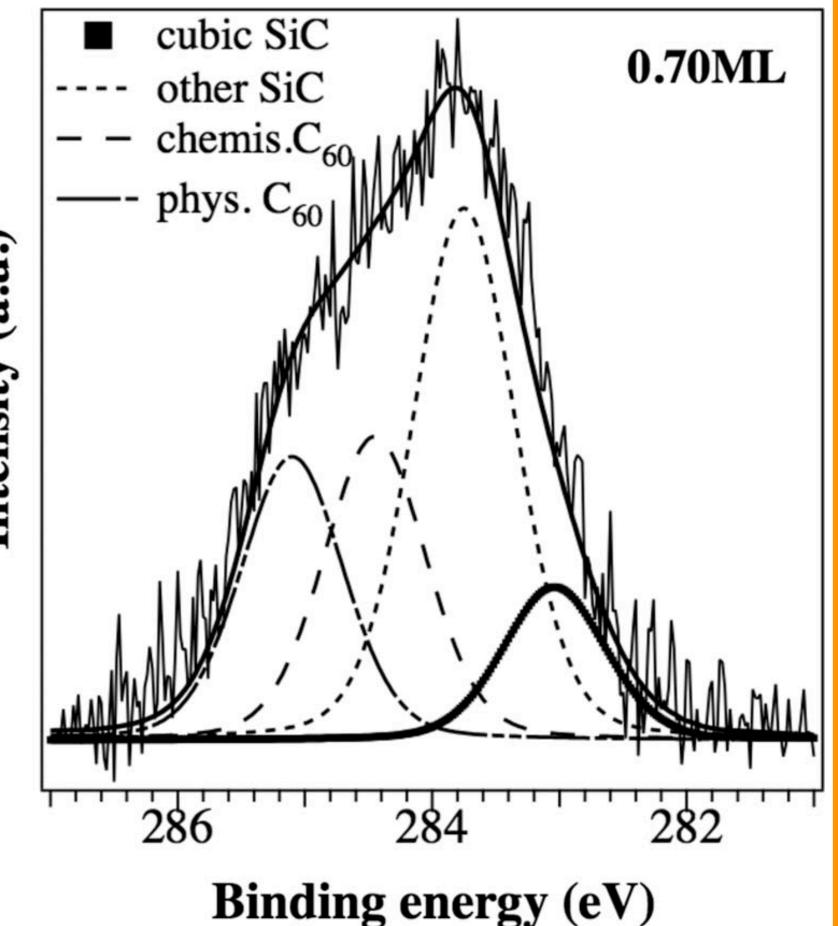






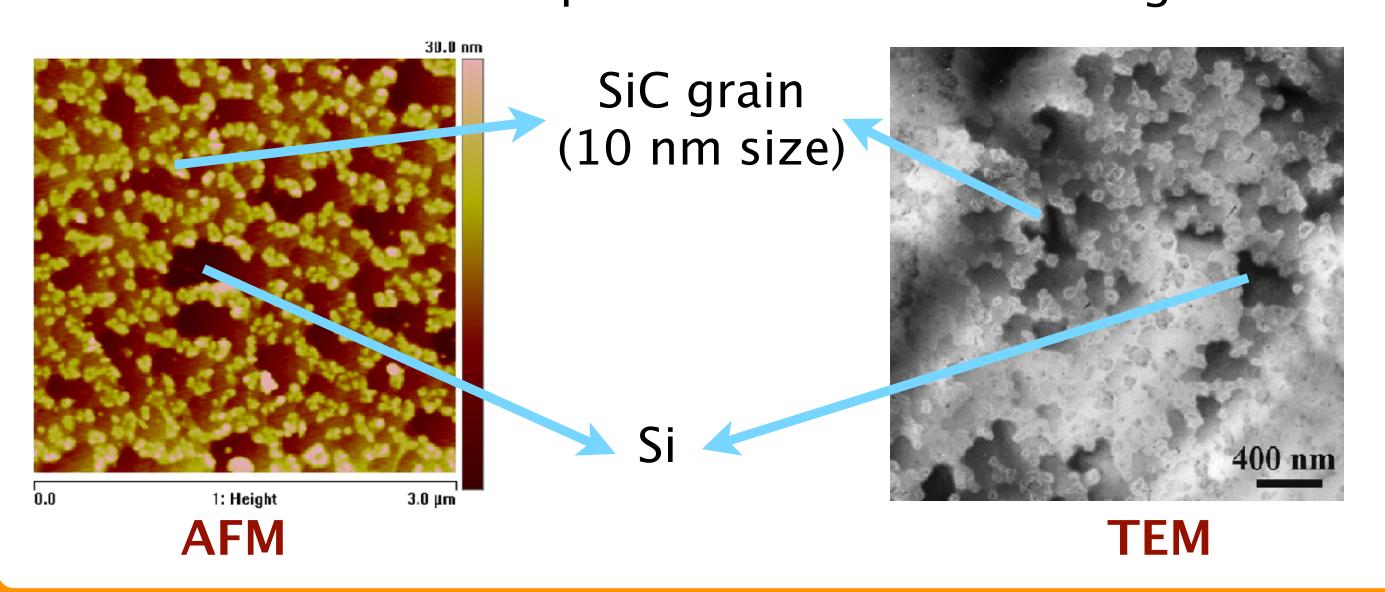
50 eV LEED patterns. (a)  $Si(111)7 \times 7$ .  $C_{60}/Si(111)7 \times 7$  film, (b) 0.70 ML, 35 eV KE, showing 3C-SiC  $1\times1$  extraspots. (c) Subtraction of image (a) to image (b): dark areas belong to the  $7\times7$  pattern, white spots to the new 3CSiC  $1\times1$ .





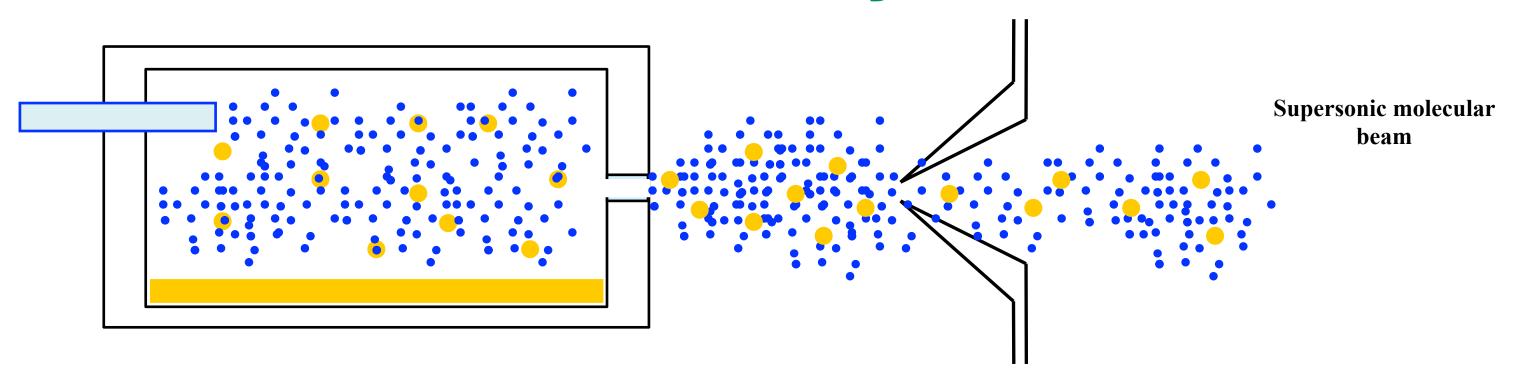
#### Valence and core photoelectron spectra

(left) Valence Bands of films grown at C<sub>60</sub> KEs of 35 eV and 30 eV, and of substrate, C60 multilayer and 3C-SiC film (right) C1s core level photoemission for a film of C60 on Si(111)–(7x7) surface at room temperature deposited by SuMBE, at a coverage of 0.7 ML. The curve has been fitted using four components, corresponding to different carbon chemical species as shown in the legend.



#### References [Experiment] R. Verucchi et al. J. Am. Chem. Soc. **134** (2012) 17400 [Theory] S. Taioli et al. J. Chem. Phys. (in press). doi:10.1063/1.4774376

# Production by SuMBE

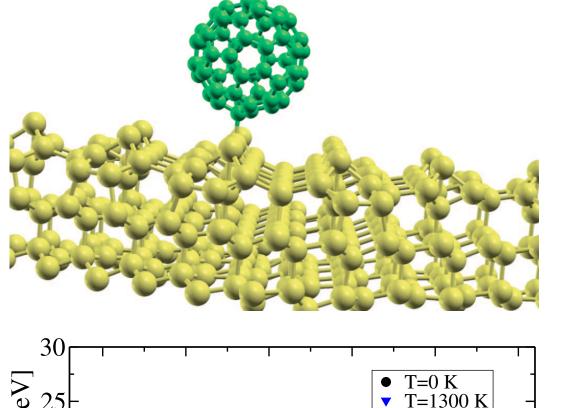


Supersonic beam produced by seeding a gas carrier (H2, He, Ar, ...) with C<sub>60</sub> out of the injection cell due to aerodynamic acceleration. Kinetic energy proportional to mass (up to tens of eV per C<sub>60</sub> molecule).

High directionality and flux, low divergence.

Experiments were performed by SuMBE deposition of C<sub>60</sub> at 30 and 35 eV onto the cleaned Si(111)-7x7 surface, successfully obtaining heteroepitaxial thin-film SiC growth at room temperature.

### **DFT-BO** simulations

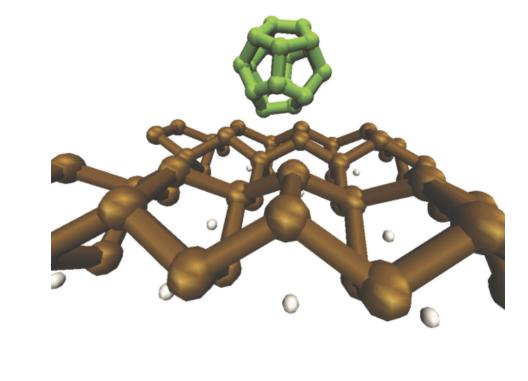


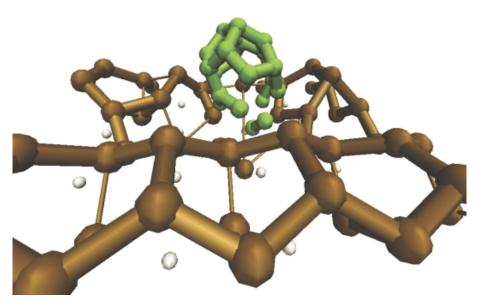
● T=0 K ▼ T=1300 K Initial kinetic energy [eV]

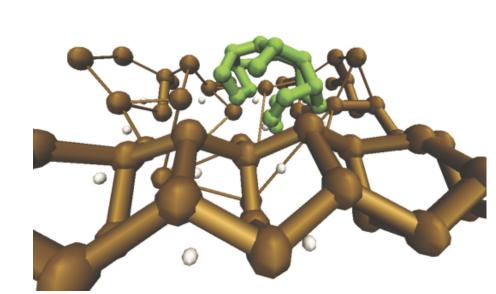
Born-Oppenheimer simulations at the level of Density Functional theory cannot reproduce C<sub>60</sub> cage breaking at the experimentally observed value of ~35 eV / molecule, although we tried: 1) increase the temperature of the substrate, 2) charge the  $C_{60}$ , 3) use  $C_{57}$ , C<sub>58</sub> or C<sub>59</sub>, 4) smash two fullerenes on top of each other.

These models predict cage breaking with KE > 300 eV.

# Non-adiabatic MD



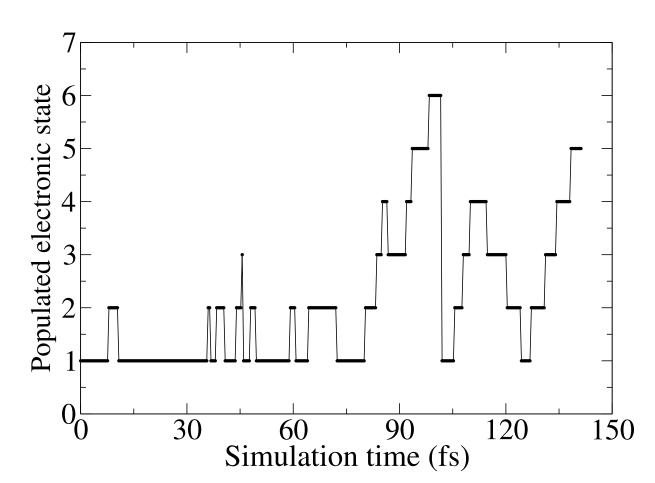




Car-Parrinello MD lowers the C<sub>60</sub> cage breaking energy down to  $\sim 100 \text{ eV} \implies \text{non-adiabatic effects can be important.}$ 

Massey parameter (~10) indicates that nuclear and electron timescales are not completely disentangled

Fully non-adiabatic simulation using TD-DFT and 6 excited states. Due to the highly intensive CPU requirements, we downscaled to C<sub>20</sub>. Cage break observed at 11 eV .



Forces used in the MD simulation are calculated on the adiabatic surfaces populated at the present time step.

Landau-Zener theory used to jump among the electronic surfaces.